



# Study of Ag/Ce<sub>x</sub>Nd<sub>1-x</sub>O<sub>2</sub> nanocubes as soot oxidation catalysts for gasoline particulate filters: Balancing catalyst activity and stability by Nd doping

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## ARTICLE INFO

### Article history:

Received 18 July 2016

Received in revised form 1 September 2016

Accepted 4 October 2016

Available online 4 October 2016

### Keywords:

Ag/Ce<sub>x</sub>Nd<sub>1-x</sub>O<sub>2</sub> nanocubes

Soot oxidation

Thermal stability

Oxygen vacancies

O<sub>x</sub><sup>-</sup> and bulk oxygen

## ABSTRACT

Three nanocubic Ag/Ce<sub>x</sub>Nd<sub>1-x</sub>O<sub>2</sub> catalysts were synthesized to evaluate the influence of Nd doping on catalyst practicability. It was observed that Nd doping generated excessive ceria surface oxygen vacancies, but reduced the bulk oxygen vacancies. These variations in oxygen vacancies suppress the utilization of ceria bulk oxygen, and induce the formation of less active O<sub>2</sub><sup>-</sup> species instead of O<sub>2</sub><sup>-</sup>, resulting in a low soot oxidation activity. However, the introduction of Nd can inhibit Ag sintering and thus enhance the catalysts' thermal stability effectively. As a consequence, Ag/Ce<sub>0.95</sub>Nd<sub>0.05</sub>O<sub>2</sub> nanocubes make a good balance between the catalyst activity and stability. Moreover, it was observed that catalyst deactivation during soot oxidation becomes more obvious when the catalysts and soot are not in close contact.

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## 1. Introduction

Gasoline direct injection (GDI) concepts are the key technology of gasoline engine development to reduce CO<sub>2</sub> emissions while improving torque and power output. However, there is a side effect of increased particle number (PN) emissions of GDI engines compared with conventional Port Fuel Injection engines [1]. One possibility to achieve the proposed Euro 6 legislation PN limit of  $6 \times 10^{11} \text{ km}^{-1}$  is the introduction of a catalyzed gasoline particulate filter (CGPF) to the exhaust after-treatment system [2]. Based on the vast experience accumulated concerning diesel vehicles, platinum catalysts seem to be the most practical ones for soot oxidation. However, the working condition of GPF is quite different from the diesel particulate filter (DPF), which is because the upstream Three-way Catalyst (TWC) consumes a large content of oxygen and converts most of NO<sub>x</sub> into N<sub>2</sub> [3,4]. Based on some earlier studies [5–8], this special working condition may provide opportunities for Ag/CeO<sub>2</sub> versus commercial platinum-based materials as soot oxi-

dation catalysts. Therefore, the application potential of Ag/CeO<sub>2</sub> for CGPF should be explored in detail.

In previous studies [9,10], it was found that monocrySTALLINE nanocubic ceria-based catalysts exhibits a high activity for soot oxidation, and the loading of silver can further improve their catalytic performance [5]. Nevertheless, the practicability of nanocubic Ag/CeO<sub>2</sub> catalysts is questionable since Ag/CeO<sub>2</sub> was reportedly exhibited severe deactivation for soot oxidation after high-temperature ageing [11–13]. Fortunately, Shimizu et al. [13] observed that lanthanum doping can increase the thermal durability of “rice-ball”-like CeO<sub>2</sub>-Ag materials, which is in agreement with the well-known knowledge that the introduction of rare-earth metals enhances the stability of ceria [6,14]. As a newly studied rare-earth metal, neodymium has proven itself good additives for ceria-based soot oxidation catalysts [15–17]. Therefore, it is worth trying to evaluate the practicability of Nd-modified nanocubic Ag/CeO<sub>2</sub> catalysts by testing their activity and thermal stability.

Furthermore, based on ceria catalysts with different shapes, we recently evidenced that excessive surface oxygen vacancies (V<sub>O-s</sub>) may lead to the formation of less active O<sub>x</sub> species, which further causes catalyst deactivation during the reaction [5,10]. The universality of this theory should be verified, both on doped ceria catalysts

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and in more practical reaction conditions. As a lanthanide dopant with positive association energies, Nd can increase the conversion rate from  $\text{Ce}^{4+}$  ionization state to  $\text{Ce}^{3+}$  state and consequently increase the oxygen vacancies in ceria nanoparticles [18]. In this sense,  $\text{Ag/CeO}_2$  doped with Nd can be ideal catalyst models for exploration of the above mechanism. What's more, the influence of high-temperature ageing on the catalyst properties may also provide useful information to improve this theory.

In the present study, three Nd-doped  $\text{Ag/CeO}_2$  catalysts with similar cube-like shapes and diverse content of oxygen vacancies were synthesized and studied. By comparing their activity and thermal stability, the influence of Nd-doping on catalyst practicability was analyzed in detail.

## 2. Experimental

### 2.1. Materials

$\text{CeO}_2$  and  $\text{Ce}_x\text{Nd}_{1-x}\text{O}_2$  nanocubes were synthesized via a hydrothermal method [5]. During a typical synthesis,  $\text{Ce}(\text{NO}_3)_3 \cdot 6\text{H}_2\text{O}$  (4 mmol),  $\text{NaOH}$  (480 mmol) and  $\text{Nd}(\text{NO}_3)_3 \cdot 6\text{H}_2\text{O}$  (0, 0.2 or 0.4 mmol) were dissolved in 80 mL of distilled water and stirred for 0.5 h. The obtained mixture was heated in a 100 mL stainless autoclave at  $180^\circ\text{C}$  for 24 h. After filtration, washing and calcinations at  $550^\circ\text{C}$  in static air for 3 h, the nanocubic samples were obtained. The supported  $\text{Ag}$  (5 wt.%) catalysts were prepared via incipient wetness impregnation of the above samples with  $\text{Ag}(\text{NO}_3)_2$  solutions (Aladdin, 99%), followed by drying at  $110^\circ\text{C}$  overnight and calcination at  $500^\circ\text{C}$  for 3 h. Then the fresh catalysts were obtained and denoted as  $\text{AgCe-F}$ ,  $\text{AgCeNd}_{0.05}\text{-F}$  ( $\text{Ag/Ce}_{0.95}\text{Nd}_{0.05}\text{O}_2$ ,  $\text{Nd:Ce} = 1:20$ ) and  $\text{AgCeNd}_{0.1}\text{-F}$  ( $\text{Ag/Ce}_{0.9}\text{Nd}_{0.1}\text{O}_2$ ,  $\text{Nd:Ce} = 1:10$ ). These catalysts were further treated at  $700^\circ\text{C}$  in 1%  $\text{O}_2/10\%$   $\text{H}_2\text{O}/\text{N}_2$  for 48 h to obtain the aged samples (labeled with “-A”). This ageing condition was chosen to simulate the working atmosphere of CGPFs [3–5] and full useful life vehicle aging (120 k miles) [19].

### 2.2. Activity measurements

As previously reported, Printex-U is a reasonable substitute for GDI soot, since they share similar morphologies and oxidation behavior [1,2]. Therefore, Printex-U from Degussa (diameter 25 nm, surface area  $100\text{ m}^2/\text{g}$ ) was used as the model soot in the present study. For each activity test, 10 mg of soot and 100 mg of catalyst were mixed by a spatula for 2 min to achieve “loose-contact” conditions. To minimize the effect of hot spots, this soot-catalyst mixture was further diluted with 300 mg of silica pellets and then sandwiched by quartz wool which placed in a tubular quartz reactor. A gas mixture of 1%  $\text{O}_2/\text{N}_2$  (500 mL/min) was fed to simulate the working condition of a CGPF roughly [5,10]. Two types of test conditions were applied in this work:

- Temperature-programmed oxidation (TPO) process at a constant heating rate ( $5^\circ\text{C}/\text{min}$ ) from room temperature (RT) to  $600^\circ\text{C}$ . The results can reflect the catalyst activity roughly.
- Isothermal reactions at 300, 350 and  $400^\circ\text{C}$  for 0.5 h, respectively. The results can reveal the catalytic behavior of the samples in detail.

The concentration of  $\text{CO}_2$  and  $\text{CO}$  in the outlet gas was monitored by an infrared spectrometer (Thermo Nicolet iS10). Each test was repeated two times to ensure the reproducibility of the results.

### 2.3. Characterizations

X-ray powder diffraction analysis of the products was performed using a Bragg–Brentano-type powder diffractometer (D8 ADVANCE, Bruker, Germany, operated at 40 kV and 30 mA, Cu–K $\alpha$  radiation,  $\lambda = 0.15418\text{ nm}$ ).

Sample morphologies were observed through a field emission scanning electron microscope (MERLIN VP Compact, ZEISS, Germany), a transmission electron microscopy (JEOL 2100 with an accelerating voltage of 200 kV and a point resolution of 0.19 nm) and  $\text{N}_2$  adsorption/desorption isotherms at  $-196^\circ\text{C}$  (JW-BK122F, Beijing JWGB, China). The size distribution of the catalysts was determined by measuring the edge length of  $>800$  cubic particles for each sample.

X-ray photoelectron spectra (XPS) were recorded on an ESCALAB 250 Xi system equipped with monochromatic Al K $\alpha$  (1486.6 eV) X-ray source. The binding energy of C 1s (284.8 eV) was used as an internal standard. The spectra were fitted using the XPSPEAK program by the curve fitting with the Gaussian function after subtraction of the Shirley-type background.

Raman spectra were obtained with a LabRAM HR 800 (HORIBA Jobin Yvon, France) spectrometer at room temperature and atmospheric pressure. A wavelength of 532 nm was used for the exciting source from an argon ion laser. The Laser beam was focused onto an area  $0.1 \times 0.1\text{ mm}^2$  in size of the sample surface. The wavenumber values of the Raman spectra were accurate to  $1\text{ cm}^{-1}$ .

$\text{H}_2$  temperature-programmed reduction ( $\text{H}_2$ -TPR) was carried out on a Micromeritics AutoChem II 2920. For each test, 50 mg catalyst was placed in a quartz tube and then put in contact with the reaction atmospheres. It is well known that the absorbed oxygen species generating from the reaction atmosphere (1%  $\text{O}_2/\text{N}_2$ ) is crucial for the catalytic oxidation of soot [5–8], but they are not even evolved during the traditional  $\text{H}_2$ -TPR tests. Therefore, this method can only give out the catalyst reducibility roughly. To investigate the catalyst utility of  $\text{O}_x^{n-}$  more realistically, cycled TPR was performed in this work. Details of the tests are illustrated in Fig. 1, which can be described as:

- The catalyst was first pre-oxidized in 1%  $\text{O}_2/\text{He}$  at  $300^\circ\text{C}$  for 10 min. Then a TPR test from  $-20^\circ\text{C}$  to  $300^\circ\text{C}$  was performed and labeled with “ $\text{O}_2$ -”. This was designed to investigate the  $\text{O}_x^{n-}$  generation ability of the catalysts [5,10].
- After this test, the samples were cooled to  $-20^\circ\text{C}$  without changing the atmosphere (10%  $\text{H}_2/\text{Ar}$ ), another TPR was then performed and labeled with “ $\text{H}_2$ -”. According to our previous study [5], the surface oxygen species observed in the “ $\text{H}_2$ -” tests can only be produced via the migration of ceria bulk oxygen to catalyst surface. This is because the catalyst surface oxygen has already been consumed by the previous “ $\text{O}_2$ -” tests, and no gaseous oxygen was introduced before the “ $\text{H}_2$ -” tests. In this sense, results of the “ $\text{H}_2$ -” tests can reveal the availability of ceria bulk oxygen over different catalysts.
- The above processes were repeated three times to test the redox stability of the catalysts, including the regeneration ability of  $\text{O}_x^{n-}$  and the utility of bulk oxygen in redox cycles. The results were denoted with “-1st”, “-2nd” and “-3rd”, respectively.

## 3. Results

### 3.1. Solid properties

Fig. 2 shows the X-ray diffractograms of the fresh and aged catalysts. All of them exhibit high crystallinity and a typical cubic fluoride  $\text{CeO}_2$  crystal phase. No peak splitting can be detected for the Nd-containing catalysts, indicating the formation of solid solu-

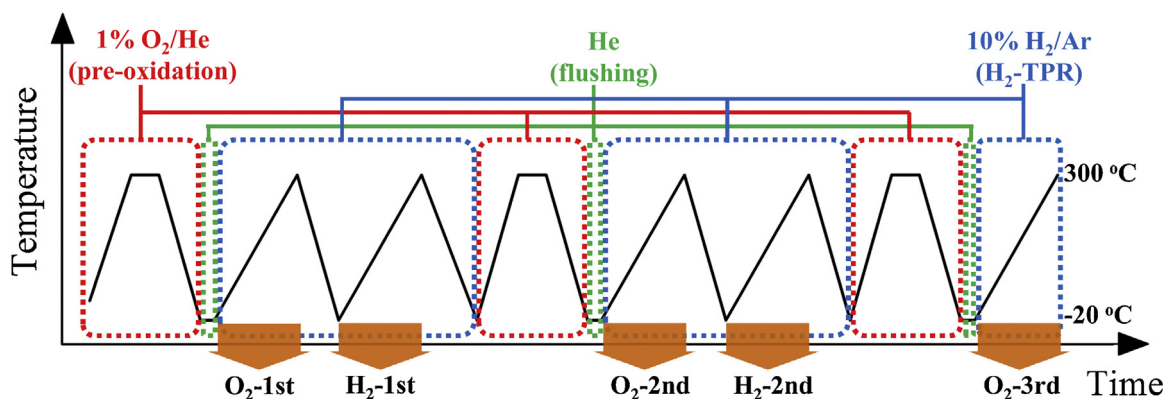


Fig. 1. Scheme of the experiments procedure of cyclic TPR.

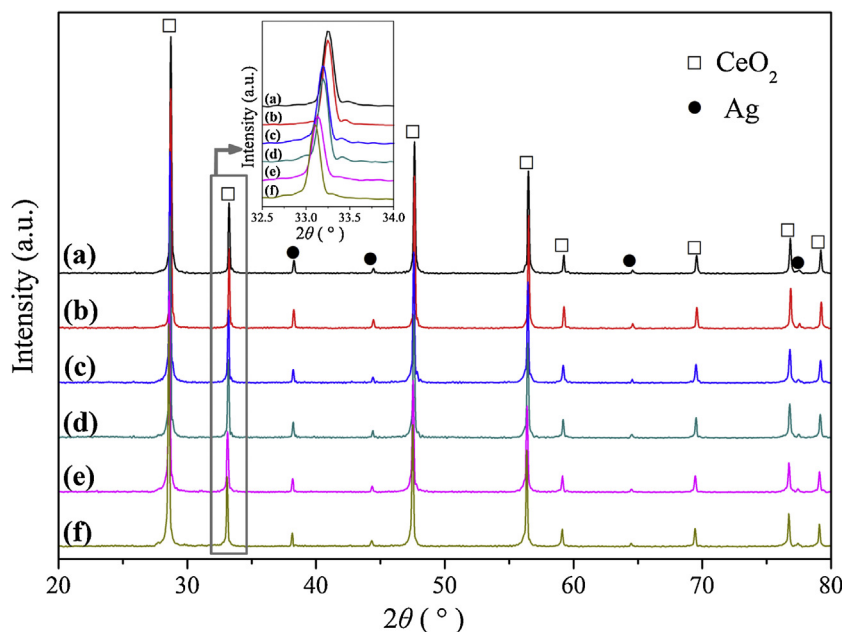


Fig. 2. XRD patterns of (a) AgCe-F, (b) AgCe-A, (c) AgCeNd<sub>0.05</sub>-F, (d) AgCeNd<sub>0.05</sub>-A, (e) AgCeNd<sub>0.1</sub>-F and (f) AgCeNd<sub>0.1</sub>-A.

tions between CeO<sub>2</sub> and Nd [15–17]. Moreover, the peak positions shift toward lower  $2\theta$  values with increasing Nd content, which should be attributed to the mismatch in ionic radius of Ce<sup>4+</sup> (97 pm) and Nd<sup>3+</sup> (112 pm). Notably, no obvious changes in peak wide and intensity are observed after ageing. This is because the fresh catalysts already exhibit high crystallinity and large crystallite sizes (see the following SEM and TEM data), then the high temperature ageing results in mainly shape transformation rather than sintering of the grains [20].

In addition to the CeO<sub>2</sub>-derived peaks, the signals at  $2\theta = 38.2^\circ$ ,  $44.3^\circ$ ,  $64.4^\circ$  and  $77.4^\circ$  suggest formation of metallic silver. No evidence for the presence of Ag<sub>2</sub>O can be obtained, both before and after ageing, which is in agreement with the results obtained by Castoldi [8], Aneggi [11], Shimizu [12] and Kayama [13] et al.

The morphology and structure of the catalysts were further explored with SEM and TEM. As shown in Fig. 3, the fresh catalysts all exhibit nanocubic morphologies with similar edge length distributions (50–400 nm). Moreover, highly dispersed metallic Ag particles (<5 nm) can be observed and evidenced by a lattice fringe of 0.24 nm (Ag{111}, see Fig. 3c, f, and i), while no AgO<sub>x</sub> species were detected. The particle size distributions of silver agree well with the results obtained by Hu et al. [21], in which silver deposited on dif-

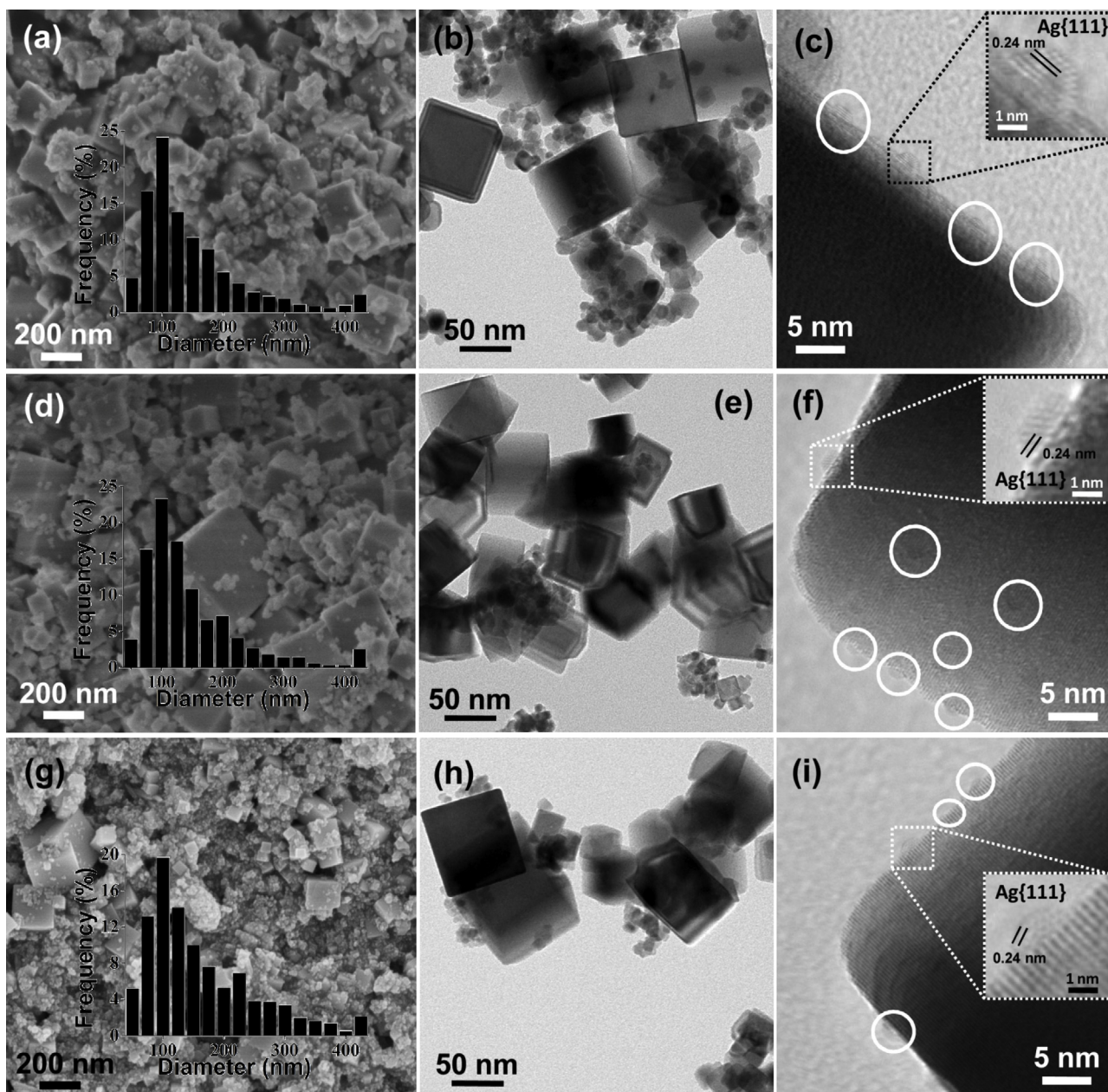
ferent CeO<sub>2-x</sub>(111) thin films maintains as nanoparticles (<5 nm) at temperatures lower than 527 °C (800 K).

After ageing treatment, the geometries of AgCe-A changed from cubes to truncated cubes (and sometimes nearly spheres, see Fig. 4a and b), which agrees well with the results obtained by Aneggi et al. [20]. It is interesting to note that AgCeNd<sub>0.05</sub>-A and AgCeNd<sub>0.1</sub>-A are still composed of sharp-faced nanocubes, indicating the doping of Nd helps maintain the initial morphologies of the catalysts. Anyhow, when comparing Fig. 3a, d and g with Fig. 4a, d and g, respectively, it is clear that the fresh and aged catalysts share similar particle size distributions. As a consequence, it is reasonable to suggest that all the catalysts exhibit similar catalyst-soot contact conditions when mixed with soot [5].

Separately, as shown in Fig. 4b and c, Ag particles with diameter of >20 nm can be detected over AgCe-A easily [12], while large Ag particles were rarely observed on AgCeNd<sub>0.05</sub>-A (Fig. 4f, g and h). As for AgCeNd<sub>0.1</sub>-A, only highly dispersed Ag nanoparticles (<5 nm) can be found (Fig. 4k and l). These results imply an increased stability of Ag nanoparticles caused by Nd doping. Similar results have also been obtained by Hu et al. dealing with Ag growth behavior over Zr-doped Ag/CeO<sub>2</sub> thin films [22].

Textural properties of the catalysts are given in Table 1. All the fresh catalysts exhibit similar surface areas and low pore volumes





**Fig. 3.** Typical SEM, TEM and HRTEM images of (a–c) AgCe-F, (d–f) AgCeNd<sub>0.05</sub>-F and (g–i) AgCeNd<sub>0.1</sub>-F.

**Table 1**  
Textural properties of the catalysts.

Catalyst	$S_{\text{BET}}$ (m <sup>2</sup> /g) <sup>a</sup>	Total pore volume (cm <sup>3</sup> /g) <sup>a</sup>	Mesopore volume (cm <sup>3</sup> /g) <sup>a</sup>	Surface Nd/Ce ratio <sup>b</sup>	Surface Ce <sup>3+</sup> /Ce <sup>4+</sup> ratio <sup>b</sup>	$I_{\text{D}}/I_{\text{F2g}}$ <sup>c</sup>
AgCe-F	17.4	0.088	0.082	–	0.22	0.12
AgCeNd <sub>0.05</sub> -F	19.1	0.117	0.111	0.37	0.27	0.11
AgCeNd <sub>0.1</sub> -F	18.9	0.118	0.112	0.64	0.34	0.04
AgCe-A	8.8	0.036	0.033	–	0.21	0.06
AgCeNd <sub>0.05</sub> -A	16.9	0.085	0.080	0.42	0.24	0.06
AgCeNd <sub>0.1</sub> -A	26.6	0.130	0.124	0.70	0.30	0.06

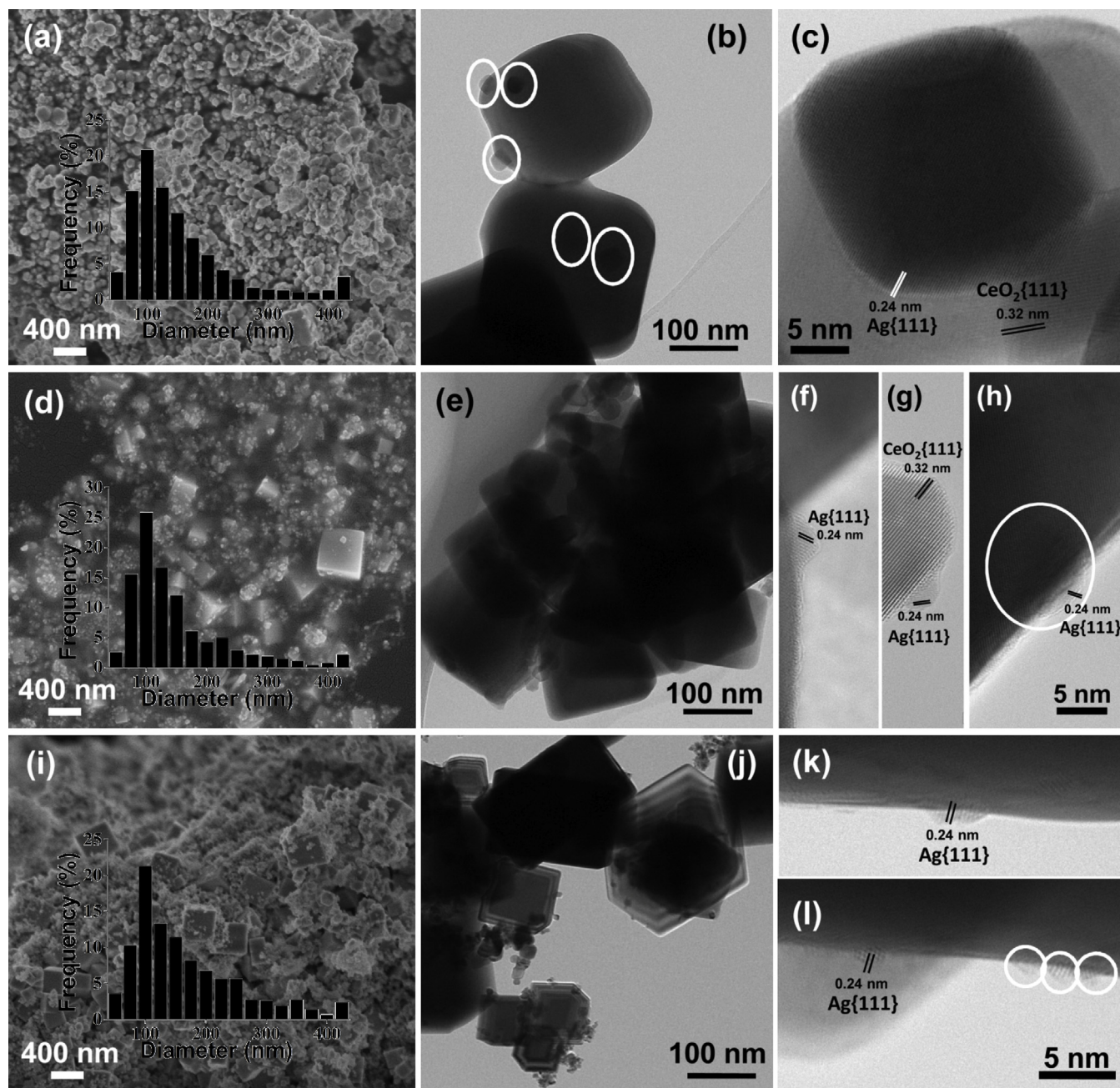
<sup>a</sup> Obtained from the nitrogen physisorption at –196 °C.

<sup>b</sup> Obtained from the XPS data.

<sup>c</sup> Obtained from the Raman data.

(see Fig. 5, note the y-scale), indicating their nonporous structure as previously reported [5,10]. After ageing, AgCe-A experienced severe loss of its surface area, while AgCeNd<sub>0.05</sub>-A and AgCeNd<sub>0.1</sub>-A show no lower  $S_{\text{BET}}$  than the corresponding fresh catalysts. This

can be explained by the extra mesopores that generated over these two samples (2–4 nm and 3–10 nm in size for AgCeNd<sub>0.05</sub>-A and AgCeNd<sub>0.1</sub>-A, respectively), which were observed distinctly by N<sub>2</sub> adsorption/desorption isotherms and TEM (see Fig. 5 and Fig. S1 in



**Fig. 4.** Typical SEM, TEM and HRTEM images of (a–c) AgCe-A, (d–h) AgCeNd<sub>0.05</sub>-A and (i–l) AgCeNd<sub>0.1</sub>-A.

the Supporting information). Similar ceria mesopores generation after ageing has also been reported by Aneggi et al. [20]. Notwithstanding, since the soot particles used in this work are far larger (~25 nm in size) than these mesopores, the above changes in  $S_{\text{BET}}$  can hardly cause crucial influence on performance of the catalysts [5].

### 3.2. Catalyst chemical states and oxygen vacancies

In order to obtain the information of Ag chemical states and CeO<sub>2</sub> surface oxygen vacancies ( $\text{V}_{\text{O-s}}$ ), XPS was performed to monitor the evolution of Ag 3d and Ce 3d spectra, the corresponding results are shown in Fig. 6 and Table 1. Some of the spectra in Fig. 6 have been normalized in intensity for a better comparison in the shapes and intensities of the peaks.

As shown in Fig. 6a, the Ag 3d<sub>5/2</sub> binding energies of the fresh samples are at ~368.1 eV. The spectra show high symmetry and a splitting of 6 eV, implying the presence of metallic Ag [23–26]. Notably, there were controversial reports about the presence of AgO<sub>x</sub> on CeO<sub>2</sub>. For example, with the assistance of HRTEM, Castoldi and Aneggi et al. [8,11] indicated that both Ag metal particle and Ag<sub>2</sub>O were present on 5 wt.% Ag/CeO<sub>2</sub>. Nevertheless, the metallic nature of silver over Ag/CeO<sub>2</sub> has also been verified by many other studies via XPS, HRTEM and EXAFS [5,21–26]. As previously reported, the ceria-based supports can accelerate the decomposition of Ag<sub>2</sub>O [27] and provide electron interaction ( $\text{Ag}^+ + \text{Ce}^{3+} \rightarrow \text{Ag}^0 + \text{Ce}^{4+}$ ) [5,23] during catalyst synthesis, which may benefit the formation of Ag<sup>0</sup> rather than AgO<sub>x</sub>. In the present work, since no evidence (XRD, HRTEM and XPS) supports the existence of AgO<sub>x</sub>, it is suggested that metallic Ag is the dominating silver species over the fresh catalysts.



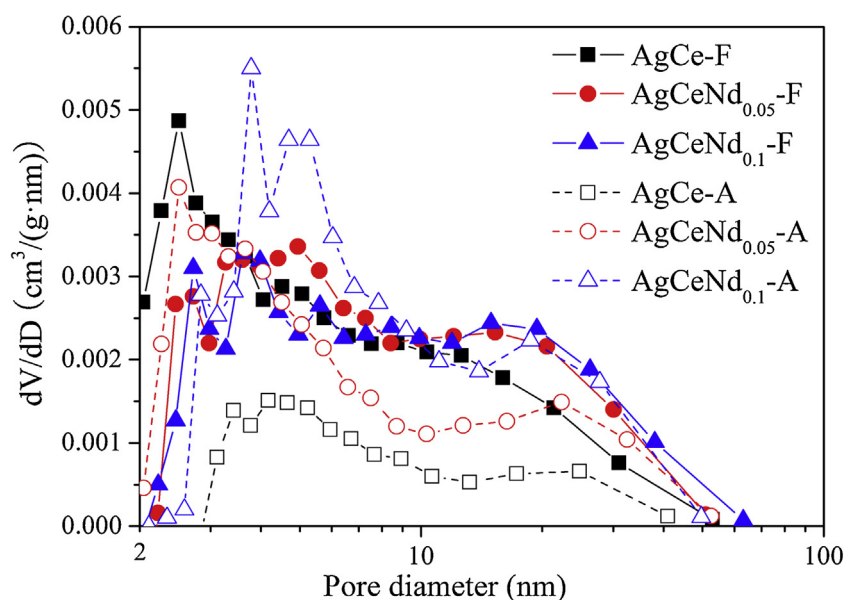


Fig. 5. Pore size distribution of the catalysts.

After annealing with 1% O<sub>2</sub> at 700 °C, the peaks attributed to Ag 3d<sub>5/2</sub> and Ag 3d<sub>3/2</sub> of Ag<sub>2</sub>O emerged at 367.9 and 373.9 eV, respectively (Fig. 6b) [24,26], indicating that the Ag particles are partially oxidized during ageing. Based on peak deconvolution, AgCe-A exhibits slightly higher content of Ag<sub>2</sub>O (Ag<sup>+</sup>/Ag<sup>0</sup> = 0.64) than AgCeNd<sub>0.05</sub>-A (0.59) and AgCeNd<sub>0.1</sub>-A (0.57). However, these Ag<sub>2</sub>O species can be detected by neither XRD nor HRTEM. These results imply the formation of ultra thin Ag<sub>2</sub>O layers on the surface of Ag nanoparticles [28,29]. Since these Ag<sub>2</sub>O species are amorphous and in small quantity, they can be detected only by surface-sensitive technique like XPS.

The surface compositions of the catalysts are given in Table 1. It is observed that the contents of surface Nd is higher than the nominal loading contents (Nd/Ce = 0.05 and 0.1 for AgCeNd<sub>0.05</sub>-F and AgCeNd<sub>0.1</sub>-F, respectively), implying the enrichment of Nd on catalysts' surface region. Similar surface enrichment of rare-earth metals over ceria has also been proven by many studies dealing with Ce<sub>x</sub>RE<sub>1-x</sub>O<sub>2-δ</sub> (RE = La, Sm, Gd, Pr and Tb) solid solutions [30–32]. These Nd components further alter the surface oxygen vacancies (V<sub>O-s</sub>) of Ag/CeO<sub>2</sub>. As shown in Fig. 6c, the four satellite peaks labeled as u<sup>0</sup>, u<sup>+</sup>, v<sup>0</sup> and v<sup>+</sup> correspond to two pairs of the spin-orbit splitting doublets, which are associated with the Ce<sup>3+</sup> cations [21,22]. By comparing the peak intensity of u<sup>+</sup> and v<sup>+</sup>, it is obvious that the Ce<sup>3+</sup> content of the catalysts follows an order of AgCeNd<sub>0.1</sub>-F > AgCeNd<sub>0.1</sub>-A > AgCeNd<sub>0.05</sub>-F > AgCeNd<sub>0.05</sub>-A > AgCe-F > AgCe-A, indicating Nd-doping induces the formation of Ce<sup>3+</sup> [18], and the ageing treatment with 1% O<sub>2</sub> led to oxidation of Ce<sup>3+</sup> to Ce<sup>4+</sup> [33]. Considering the creation of V<sub>O-s</sub> is closely related with the presence of Ce<sup>3+</sup> [5,10,34], it is suggested that the concentration of V<sub>O-s</sub> follows the same sequence for the catalysts.

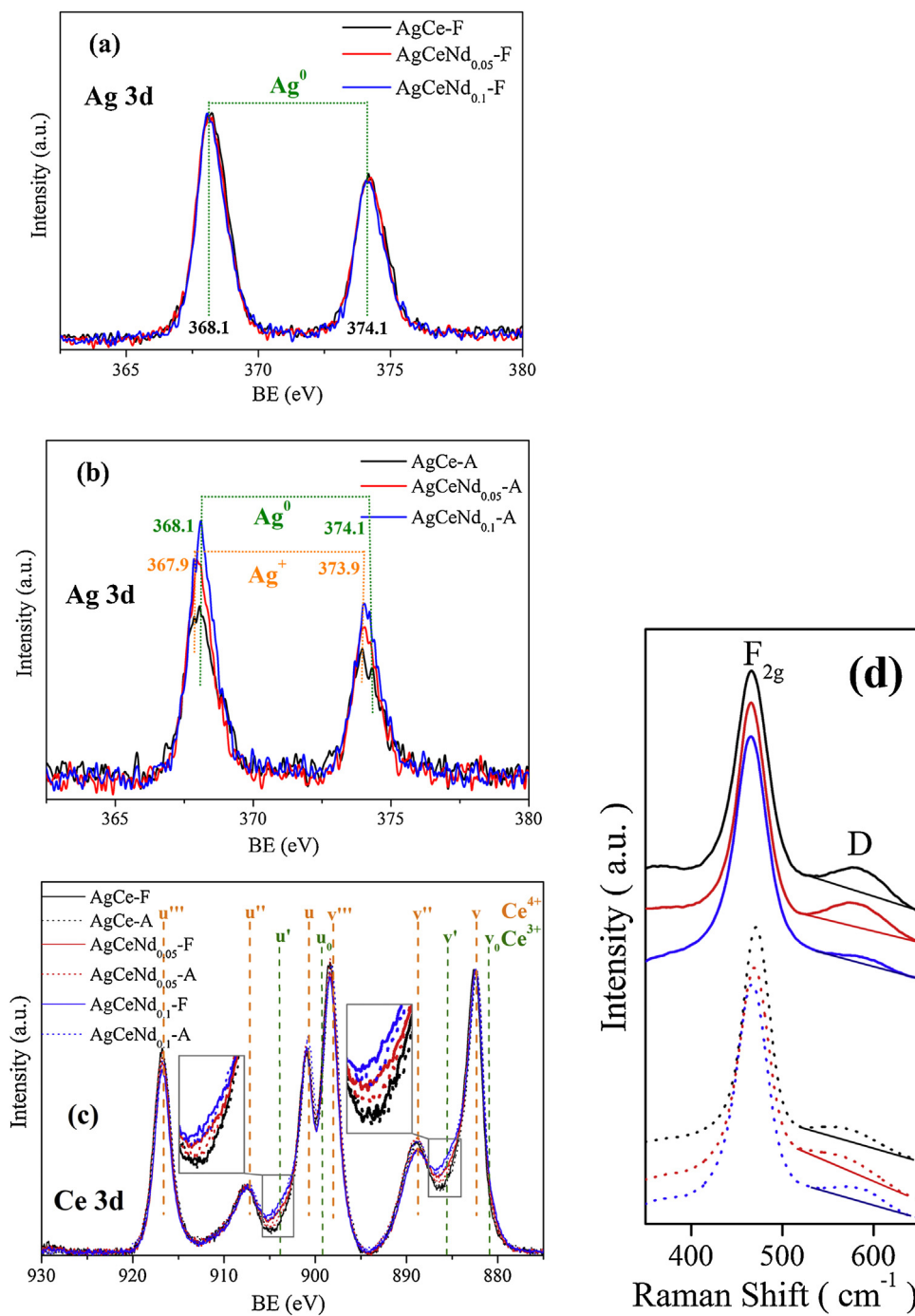
Different from the surface-sensitive XPS, visible Raman spectroscopy is a useful technique to investigate the bulk oxygen vacancies (V<sub>O-b</sub>) of the ceria-containing solid solutions [24,30–32]. As shown in Fig. 6d, all the catalysts exhibit a strong peak at ~465 cm<sup>-1</sup> with a shoulder at ~600 cm<sup>-1</sup>, corresponding to the F<sub>2g</sub> and defect-induced (D) modes of cubic CeO<sub>2</sub> fluoride phase, respectively [24]. Their intensity ratios (I<sub>D</sub>/I<sub>F2g</sub>) in Table 1 indicate that the loading of 10% Nd reduced the V<sub>O-b</sub> concentration of Ag/CeO<sub>2</sub> obviously. This means Nd doping affects the creation of V<sub>O-b</sub> in a different way to that of V<sub>O-s</sub>. Possible explanations are as follows: during catalyst synthesis, Ag species over ceria leads to

oxygen reverse spillover [5,21,22], which creates V<sub>O-s</sub> that re-filled rapidly by bulk oxygen of CeO<sub>2</sub> via diffusion. Consequently, Ag promotes the generation of V<sub>O-b</sub> rather than V<sub>O-s</sub> [24]. With the loading of Nd, excessive V<sub>O-s</sub> increases the adsorption energies of O-Ce, and hereby makes oxygen reverse spillover from CeO<sub>2</sub> to silver difficult [35]. Therefore, Ag/Ce<sub>x</sub>Nd<sub>1-x</sub>O<sub>2</sub> cannot generate V<sub>O-b</sub> as readily as Ag/CeO<sub>2</sub>. A similar “obstruction” effect of Pr doping on Pd-Ce interaction and oxygen reverse spillover has been reported by Ran et al. [36].

Separately, according to the results obtained by Chueh et al. [37], bulk Ce<sup>3+</sup> can be oxidized by gaseous O<sub>2</sub> even more easily than surface Ce<sup>3+</sup>. Therefore, ageing treatment in 1% O<sub>2</sub> reconciled the V<sub>O-b</sub> concentration of all the catalysts to a similar low level (Table 1).

### 3.3. Catalytic activities for soot oxidation

Fig. 7 illustrates the catalytic performance of the samples, all the catalysts yield a high CO<sub>2</sub> selectivity during the reactions (CO<sub>2</sub>/CO<sub>x</sub> > 99%). As shown in Fig. 7a, T<sub>max</sub> (maximal soot oxidation rate temperature) of the catalysts follows an order of AgCe-F (394 °C) < AgCeNd<sub>0.05</sub>-F (413 °C) < AgCeNd<sub>0.05</sub>-A (439 °C) < AgCe-A (448 °C) < AgCeNd<sub>0.1</sub>-F (455 °C) ≈ AgCeNd<sub>0.1</sub>-A (458 °C), suggesting that the introduction of Nd decreases the activity of Ag/CeO<sub>2</sub>, but increases its thermal stability instead. Notably, the sharp CO<sub>2</sub> peak of AgCe-A indicates thermal runaway over this catalyst, which may lead to inaccurate presentation of the catalyst soot oxidation activity [38]. Therefore, it is necessary to re-evaluate the catalyst performance via isothermal reactions. Moreover, considering there is always a certain amount of water vapor in the exhaust of GPF [3], the influence of water to the catalytic performance of AgCe-F was evaluated by introducing 10% H<sub>2</sub>O during soot-TPO test. As shown in Fig. S2 in the Supporting information, water seems to benefit the oxidation of soot slightly, which has once been attributed to the elimination of adsorbed hydrogen by oxygen species and the increase in contact area caused by gasification of carbon with water [39]. Nevertheless, since these changes in catalyst activity are rather small (especially at temperatures below 320 °C), and most of the important characterizations applied in this study (e.g. TEM, XPS, Raman and especially H<sub>2</sub>-TPR) cannot be performed in a humid atmosphere, thus the following results and discussion are made based on soot oxidation reactions in absence of water.



**Fig. 6.** (a–c) XPS and (d) Visible Raman spectra of the catalysts in the (a,b) Ag 3d and (c) Ce 3d core level regions. Fig. 6c and d share the same legend.

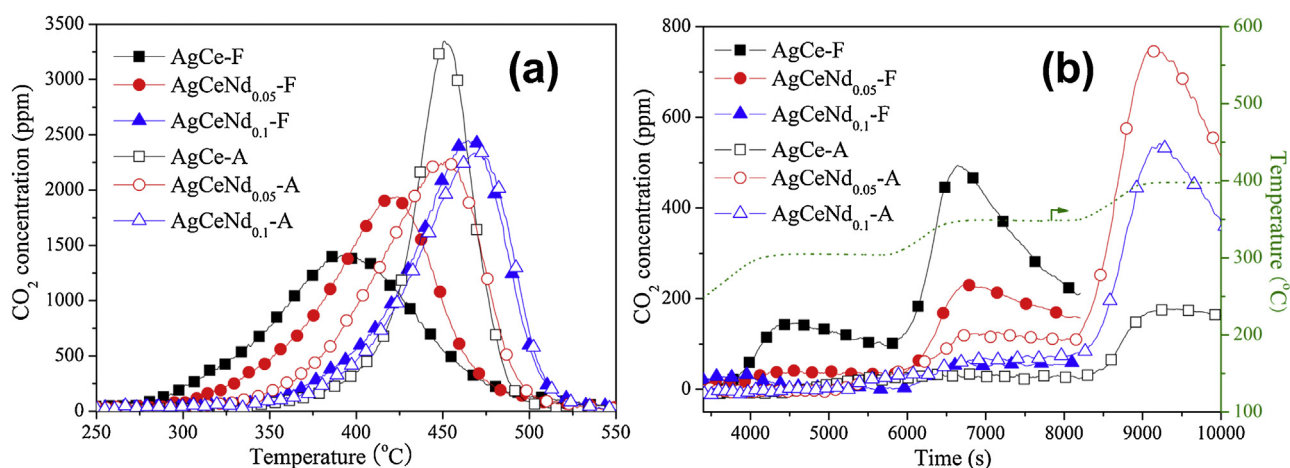
The results of isothermal soot oxidation were shown in Fig. 7b. It is noted that, since soot conversion of AgCe-F and AgCeNd<sub>0.05</sub>-F at 350 °C is relatively high (>10%), not only catalyst deactivation, but also changes in soot remaining amount may cause the drop in CO<sub>2</sub> production over these catalysts. Meanwhile, the reactions at 300 °C were in a more stable state, which could thus reflect the behavior of the catalysts more accurately. Similar attention should be paid when analyze the catalytic performance of all the aged catalysts at 400 °C.

Among the fresh catalysts, AgCe-F exhibits the highest reaction rate, following by AgCeNd<sub>0.05</sub>-F and AgCeNd<sub>0.1</sub>-F. All of them exhibited time on-stream deactivation, which was attributed to insufficient O<sub>x</sub><sup>n-</sup> supplement in our previous works [5,10]. It is noteworthy that, no deactivation was observed over AgCe-F when

the reactions are performed in “tight-contact” conditions (not shown), indicating the lack of catalyst-soot contact points in “loose-contact” conditions makes O<sub>x</sub><sup>n-</sup> supplement even more difficult. After ageing, the catalysts are almost inert at 300 °C. When the temperature was raised to 350 and 400 °C, AgCeNd<sub>0.05</sub>-A followed by AgCeNd<sub>0.1</sub>-A exhibits obviously better activities than AgCe-A. Interestingly, AgCeNd<sub>0.1</sub>-A shows no worse catalytic performance than AgCeNd<sub>0.1</sub>-F at 350 °C, which again confirms that Nd doping improves the stability of Ag/CeO<sub>2</sub>.

#### 3.4. Cycled H<sub>2</sub> temperature-programmed reduction

As reported earlier [5–10,14], surface oxygen species (O<sub>x</sub><sup>n-</sup>) are the main active phases for soot oxidation over ceria-based catalysts,



**Fig. 7.** CO<sub>2</sub> concentration during (a) temperature-programmed and (b) isothermal soot oxidation reactions. Reaction conditions: 1% O<sub>2</sub>/N<sub>2</sub> (500 mL/min), catalyst/soot = 10/1, loose contact, heating rate = 5 °C/min.

and H<sub>2</sub>-TPR is an efficient method to investigate the catalyst utility of O<sub>x</sub><sup>n−</sup>. With the assistance of silver, these species can be consumed by H<sub>2</sub> at <300 °C (see Fig. S3 in the Supporting information) [5]. Therefore, each cycled TPR test was terminated at 300 °C to ensure that most of O<sub>x</sub><sup>n−</sup> can be consumed, while no obvious structural changes of the catalysts are expected since all the catalysts were synthesized via calcinations at ≥500 °C.

As illustrated in Fig. 8a, the H<sub>2</sub> consumption peaks at 0–40 °C, 50–70 °C and 100–240 °C over AgCe-F become distinct after pre-oxidation (comparing with the profiles in Fig. S3). According to our previous studies [5,23], these peaks can be assigned to reduction of surface-adsorbed O<sub>2</sub><sup>−</sup>, O<sup>−</sup> and surface capping oxygen (O<sup>2−</sup>), respectively. After carefully dealing with the profiles' baselines (see Fig. S4a in the Supporting information for details), quantitative data were obtained and shown in Table 2. Based on the discussion in Section 2.3, the cycled H<sub>2</sub>-TPR results can be analyzed in three parts: (1) the “O<sub>2</sub><sup>−</sup>” data concerning about catalyst O<sub>x</sub><sup>n−</sup> generation ability, (2) the “H<sub>2</sub>” data concerning about catalyst availability of ceria bulk oxygen and (3) differences between the “-1st”, “-2nd” and “-3rd” data, which reflect catalyst redox stability and O<sub>x</sub><sup>n−</sup> regeneration.

(1) For the fresh catalysts, AgCeNd<sub>0.05</sub>-F and AgCeNd<sub>0.1</sub>-F exhibit significantly lower amount of O<sub>2</sub><sup>−</sup> than AgCe-F in the “O<sub>2</sub><sup>−</sup>” tests (see Fig. 8b, c and Table 2). Their O<sup>−</sup> and O<sup>2−</sup>-derived H<sub>2</sub> consumption peaks become overlapped, and the proportion of O<sup>2−</sup> goes up remarkably with Nd content. These results were reconfirmed by Raman (see Fig. S4b), indicating excessive V<sub>O-s</sub> induces the formation of less active oxygen species (like O<sup>2−</sup>) rather than O<sub>2</sub><sup>−</sup> [5,10]. After ageing, the amount of O<sub>x</sub><sup>n−</sup> (especially O<sub>2</sub><sup>−</sup>) generated over AgCe-A was reduced drastically compared with AgCe-F (Fig. 8d and Table 2). Meanwhile, AgCeNd<sub>0.05</sub>-A and AgCeNd<sub>0.1</sub>-A exhibit even higher content of O<sub>2</sub><sup>−</sup> than the corresponding fresh catalysts (see Fig. 8e, f and Table 2). Considering O<sub>2</sub><sup>−</sup> is highly active for soot catalytic oxidation [5–7,13], these results explain why ageing treatment caused less impact on the Nd-containing catalysts.

Notably, H<sub>2</sub> consumption occurred at higher temperatures over AgCeNd<sub>0.05</sub>-A and especially AgCe-A compared with the fresh catalysts (comparing Fig. 8a and b with Fig. 8d and e, respectively). This is because sintering of the Ag nanoparticles decreased their active surface area for H<sub>2</sub> adsorption and spillover. Separately, AgCeNd<sub>0.1</sub>-A and AgCeNd<sub>0.1</sub>-F show similar H<sub>2</sub> reduction temperatures, which reconfirms the Ag stabilization effect caused by Nd doping.

(2) As indicated in Section 2.3, the surface oxygen species observed in H<sub>2</sub>-1st and 2nd TPRs of AgCe-F can only be produced via the migration of ceria bulk (mainly subsurface [40]) oxygen to catalyst surface. This process was proven accelerated by the partic-

ipant of silver through a reverse spillover effect [5,21,22]. As shown in Table 2, the amount of bulk oxygen-derived O<sub>x</sub><sup>n−</sup> species (H<sub>2</sub>-1st) can be comparable to the gaseous O<sub>2</sub>-derived O<sub>x</sub><sup>n−</sup> (O<sub>2</sub>-1st). In this sense, the bulk-to-surface migration of oxygen plays important role during soot oxidation over AgCe-F.

Meanwhile, as evidenced by the H<sub>2</sub>-1st and -2nd results, the migration of bulk oxygen to catalyst surface was inhibited by Nd doping (especially for AgCeNd<sub>0.1</sub>-F, see Fig. 8c). This is mainly because the loading of Nd reduces V<sub>O-b</sub> of CeO<sub>2</sub> (see Table 1 and Fig. 6d), which hereby decreases the oxygen diffusion rate in ceria bulk [34]. Moreover, since excessive V<sub>O-s</sub> inhibits the oxygen reverse spillover from CeO<sub>2</sub> to Ag [35], AgCeNd<sub>0.05</sub>-F shows less availability of ceria bulk oxygen than AgCe-F (see the “H<sub>2</sub>-1st” results in Fig. 8a, b and Table 2) although they exhibit similar content of V<sub>O-b</sub> (Table 1).

After ageing, the availability of bulk oxygen over AgCe-A becomes negligible due to severe Ag sintering (Fig. 8d and Table 2). Contrarily, AgCeNd<sub>0.05</sub>-A and AgCeNd<sub>0.1</sub>-A with high thermal stability can still produce O<sub>x</sub><sup>n−</sup> with ceria bulk oxygen (Fig. 8e, f and Table 2).

(3) By comparing the O<sub>2</sub>-1st, 2nd and 3rd results of AgCe-F (Table 2), it is clear that the amount of active oxygen species (O<sub>x</sub><sup>n−</sup>) drops obviously during the redox cycles. In this sense, during soot oxidation in 1% O<sub>2</sub> at 300 °C, once the O<sub>x</sub><sup>n−</sup> species over AgCe-F are consumed by soot, they can hardly be regenerated by gaseous O<sub>2</sub> immediately (not even in minutes, as the pre-oxidation process lasts for 10 min). Therefore, it is reasonable to speculate the deficient redox stability of AgCe-F leads to its time-on-stream deactivation. As shown in Fig. 7b, 8b, 8c and 8d, similar redox instability and deactivation can also be observed over AgCeNd<sub>0.05</sub>-F, AgCeNd<sub>0.1</sub>-F and AgCe-A. Nevertheless, AgCeNd<sub>0.05</sub>-A and especially AgCeNd<sub>0.1</sub>-A exhibit even better redox stability after ageing (Table 2). This is because they have similar Ag particle sizes but lower V<sub>O-s</sub> contents compared with the fresh catalysts.

By comparing the H<sub>2</sub>-1st and 2nd results, it is observed that the bulk oxygen-derived O<sub>x</sub><sup>n−</sup> is quite stable during the redox cycles, which indicates that ceria bulk oxygen plays important role in maintaining stable catalytic cycles. Consequently, as shown in Fig. 7b (300 °C for the fresh catalysts, 350 °C for the aged ones), the catalysts first deactivated with time on stream, then reached stable activities during isothermal soot oxidation.

As a summary, Nd doping inhibits the formation of highly active surface oxygen species and thus leads to a low initial soot oxidation activity. However, the capability to maintain certain O<sub>x</sub><sup>n−</sup> generation



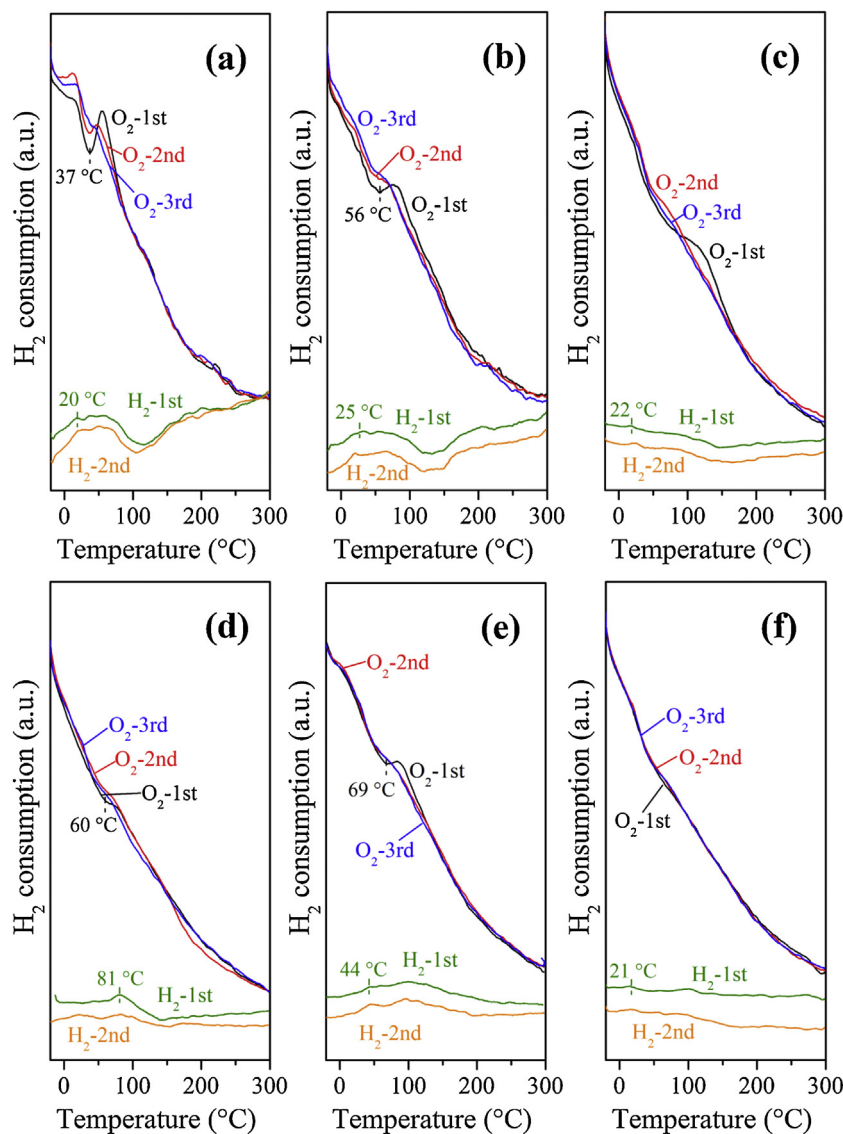


Fig. 8. Cycled TPR results of (a) AgCe-F, (b) AgCeNd<sub>0.05</sub>-F, (c) AgCeNd<sub>0.1</sub>-F, (d) AgCe-A, (e) AgCeNd<sub>0.05</sub>-A and (f) AgCeNd<sub>0.1</sub>-A.

Table 2

Quantitative analyses of the surface oxygen species-derived H<sub>2</sub> consumptions (μmol/g<sub>cat</sub>.) in cycled-TPR.

Catalyst	O <sub>2</sub> -1st			O <sub>2</sub> -2nd			O <sub>2</sub> -3rd			H <sub>2</sub> -1st		H <sub>2</sub> -2nd	
	O <sub>2</sub> <sup>2-</sup>	O <sup>-</sup>	O <sup>2-</sup>	O <sub>2</sub> <sup>2-</sup>	O <sup>-</sup>	O <sup>2-</sup>	O <sub>2</sub> <sup>2-</sup>	O <sup>-</sup>	O <sup>2-</sup>	O <sub>x</sub> <sup>n-</sup>	O <sup>2-</sup>	O <sub>x</sub> <sup>n-</sup>	O <sup>2-</sup>
AgCe-F	4.5	18.1	1.5	5.2	9.2	1.9	2.6	3.3	1.9	20.7	8.4	21.5	6.3
AgCeNd <sub>0.05</sub> -F	1.5	23.5 (o) <sup>a</sup>		1.1	14.2 (o)		0.9	11.1 (o)		17.4	5.2	19.0	3.1
AgCeNd <sub>0.1</sub> -F	0.5	27.3 (o)		0.3	9.6 (o)		0.3	5.2 (o)		5.0	1.4	4.7	1.3
AgCe-A	0	8.0 (o)		0	12.0 (o)		0.1	3.1 (o)		15.8 (o)		15.1 (o)	
AgCeNd <sub>0.05</sub> -A	2.1	23.1 (o)		2.2	10.1 (o)		1.7	9.4 (o)		17.9 (o)		18.4 (o)	
AgCeNd <sub>0.1</sub> -A	0.5	3.6 (o)		0.6	5.1 (o)		0.5	5.1 (o)		5.8 (o)		6.2 (o)	

<sup>a</sup> (o) means these peaks overlapped.

(from gaseous O<sub>2</sub> or ceria bulk oxygen) even after high-temperature ageing makes Ag/Ce<sub>x</sub>Nd<sub>1-x</sub>O<sub>2</sub> more thermally stable than Ag/CeO<sub>2</sub>.

#### 4. Discussion

In reviewing of soot oxidation mechanisms over ceria-based catalysts, surface oxygen species (O<sub>x</sub><sup>n-</sup>) have been proven as the most important active phases [5–10,14]. During a typical reaction, once these species were consumed by soot, they should be regenerated with gaseous O<sub>2</sub> and/or ceria bulk oxygen, and thus maintain a

stable catalytic cycle (a Mars-van-Krevelen-like mechanism). For Ag/CeO<sub>2</sub>, there are mainly three factors that may influence the catalytic cycle. (1) Catalyst morphologies can alter the number of catalyst-soot contact points, and then determine the transfer efficiency of O<sub>x</sub><sup>n-</sup> from catalyst to soot [5,9,40]. (2) The particle size and chemical states of Ag are closely related with both the dissociation of gaseous O<sub>2</sub> and utility of ceria bulk oxygen [41], which are crucial for the generation of O<sub>x</sub><sup>n-</sup>. (3) The concentration of oxygen vacancies (both V<sub>O-s</sub> and V<sub>O-b</sub>) was proven associated with the amount and specific type of O<sub>x</sub><sup>n-</sup> (O<sub>2</sub><sup>2-</sup>, O<sup>-</sup> and/or O<sup>2-</sup>) [5,7,10],

while the interaction between oxygen vacancies and Ag may further impact the catalytic and thermal behaviors of Ag [21,22,40]. In this sense, the following discussions were made upon (1) how Nd doping influences the above three factors, and (2) how the influenced factors further affect the entire catalytic cycle and catalyst activity. The fresh and aged catalysts were discussed separately for comparison.

#### 4.1. Effect of Nd doping on the fresh catalysts

As evidenced by HRTEM, XPS and SEM, the three fresh catalysts exhibit similar catalyst morphologies (nanocubes, see Fig. 3), Ag particle sizes (<5 nm, see Fig. 3) and chemical states ( $\text{Ag}^0$ , see Fig. 6b). Meanwhile, as evidenced by XRD (Fig. 2), XPS (Fig. 6c) and Raman (Fig. 6d), the substitution of  $\text{CeO}_2$  with trivalent  $\text{Nd}^{3+}$  ions results in formation of ceria-based solid solutions accompanied with generation of more  $\text{V}_{\text{O-s}}$  and less  $\text{V}_{\text{O-b}}$ . Therefore, factor (3) (the concentration of oxygen vacancies) domains the differences between AgCe-F, AgCeNd<sub>0.05</sub>-F and AgCeNd<sub>0.1</sub>-F. The influences of  $\text{V}_{\text{O-s}}$  and  $\text{V}_{\text{O-b}}$  on  $\text{O}_x^{\text{n-}}$  should thus be paid close attention to.

On one hand,  $\text{V}_{\text{O-s}}$  is closely related with the gaseous oxygen-derived  $\text{O}_x^{\text{n-}}$ . Over Ag/ $\text{CeO}_2$ ,  $\text{O}_2$  is transformed into  $\text{O}_x^{\text{n-}}$  via:  $\text{O}_2(\text{g}) \rightarrow 2\text{O}(+\text{e}) \rightarrow \text{O}_2^-(+\text{e}) \rightarrow 2\text{O}^{2-}(+\text{e}) \rightarrow 2\text{O}^{2-}$  [5,7,23,35]. During the first adsorption and dissociation processes ( $\text{O}_2 \rightarrow 2\text{O}$ ),  $\text{V}_{\text{O-s}}$  favors the formation of atomic O. This has been evidenced by first-principles calculations [35] and the  $\text{O}_2$ -1st TPR results in this work, in which AgCeNd<sub>0.1</sub>-F exhibits higher surface oxygen-derived  $\text{H}_2$  consumption (27.8  $\mu\text{mol/g}_{\text{catal}}$ ) than AgCeNd<sub>0.05</sub>-F (25.0  $\mu\text{mol/g}_{\text{catal}}$ ) and AgCe-F (24.1  $\mu\text{mol/g}_{\text{catal}}$ ). However, during the subsequent transformation processes ( $2\text{O} \rightarrow \text{O}_2^- \rightarrow 2\text{O}^- \rightarrow 2\text{O}^{2-}$ ), excessive  $\text{V}_{\text{O-s}}$  and  $\text{Ce}^{3+}$  lead to deep reduction of the oxygen. As a consequence, although Nd doping accelerates  $\text{O}_x^{\text{n-}}$  formation, it favors the formation of  $\text{O}^{2-}$  instead of the highly active  $\text{O}_x^-$  (Table 2) [5,7].

On the other hand, both  $\text{V}_{\text{O-b}}$  and  $\text{V}_{\text{O-s}}$  are crucial for the bulk oxygen-derived  $\text{O}_x^{\text{n-}}$ . Under reduction conditions (e.g. when catalyst reacts with  $\text{H}_2$  or soot), atomic oxygen (O) can release from ceria bulk and transform into surface  $\text{O}_x^{\text{n-}}$  through:  $\text{O}^{2-}(\text{bulk}) \rightarrow 2\text{O} \rightarrow \text{O}_2^- \rightarrow 2\text{O}^- \rightarrow 2\text{O}^{2-}$  [5,34,40]. The first step ( $\text{O}^{2-}(\text{bulk}) \rightarrow 2\text{O}$ ) determines the catalyst utility of bulk oxygen, which can be improved by a high content of  $\text{V}_{\text{O-b}}$  and the reverse spillover of oxygen from  $\text{CeO}_2$  to Ag [5,21,22,34]. However, as indicated in Sections 3.2 and 3.4, due to the reduced  $\text{V}_{\text{O-b}}$  (Fig. 6d) and extra  $\text{V}_{\text{O-s}}$  (Fig. 6c) caused by Nd doping, both the steps were inhibited over Ag/ $\text{Ce}_x\text{Nd}_{1-x}\text{O}_2$ . As a consequence, the bulk oxygen in AgCeNd<sub>0.05</sub>-F and especially AgCeNd<sub>0.1</sub>-F cannot be fully utilized to generate surface  $\text{O}_x^{\text{n-}}$ , which is supported clearly by the suppressed  $\text{H}_2$  consumption during  $\text{H}_2$ -1st and -2nd TPRs of AgCeNd<sub>0.1</sub>-F (Fig. 8c).

As a summary, since the doping of Nd inhibits the availability of ceria bulk oxygen, and induces the formation of less active oxygen species as well, Ag/ $\text{Ce}_x\text{Nd}_{1-x}\text{O}_2$  can hardly provide enough active species during the reaction and regeneration processes. Therefore, it is not surprising that both AgCeNd<sub>0.05</sub>-F and AgCeNd<sub>0.1</sub>-F exhibit far lower soot oxidation activity than AgCe-F (Fig. 7b).

#### 4.2. Effect of Nd doping on the aged catalysts

To understand the influences of Nd additives on the aged catalysts, one should first be aware of the roles that Nd species played during ageing. As previously mentioned, although the morphologies of AgCe-A (cubes, truncated cubes and spheres, see Fig. 4a and b) are slightly different from the Nd-doping catalysts (cubes, see Fig. 4d, e, i and j), their identical particle size distributions imply similar catalyst-soot contact conditions [5]. In this sense, only factor (2) (Ag particle size and chemical states) and factor (3) (the con-

centration of oxygen vacancies) were concerned in the following discussions.

First, about the particle sizes of Ag: as illustrated in Table 1, Fig. 5 and Fig. S1 (see the Supporting information), both AgCeNd<sub>0.05</sub>-A and AgCeNd<sub>0.1</sub>-A generate extra mesopores during ageing, resulting in their higher  $S_{\text{BET}}$  compared with AgCe-A. Although these mesopores contribute little to soot catalytic oxidation [5], they led to isolation of the primary Ag nanoparticles and hereby hampered silver migration upon aging [42]. Furthermore, Hu et al. [21,22] have proven that surface oxygen vacancies increase the bonding of Ag to the ceria surface, which implies that the excessive  $\text{V}_{\text{O-s}}$  over Ag/ $\text{Ce}_x\text{Nd}_{1-x}\text{O}_2$  catalysts may help inhibit Ag sintering. In either case, the Ag nanoparticles over Ag/ $\text{Ce}_x\text{Nd}_{1-x}\text{O}_2$  are far more stable than those over Ag/ $\text{CeO}_2$ , which was evidenced by the HRTEM (Fig. 4) and  $\text{H}_2$ -TPR (Fig. 8) results.

The sintering of Ag can affect the catalytic oxidation of soot significantly. For example, Shimizu [12] and Haneda et al. [43] found that large Ag particles (>15 nm) supported on  $\text{CeO}_2$ ,  $\text{ZrO}_2$ ,  $\text{TiO}_2$  and MgO showed obviously worse soot oxidation activities than Ag nanoparticles (<5 nm). Their low activity has been attributed to the decrease in Ag surface area and hereby weakened  $\text{O}_2$  activation ability [41]. This was also evidenced by the  $\text{H}_2$ -TPR results in this work: as shown in Fig. 8d and Table 2, AgCe-A can hardly generate any  $\text{O}_2^-$  during  $\text{O}_2$ -1st, 2nd and 3rd TPRs. Contrarily, AgCeNd<sub>0.05</sub>-A and AgCeNd<sub>0.1</sub>-A generates no less  $\text{O}_x^-$  than the corresponding fresh catalysts, indicating the Ag nanoparticles over their surface can still work well to adsorb and activate gaseous  $\text{O}_2$ . What's more, Ag sintering leads to loss of  $\text{CeO}_2$ -Ag contact, which impacts the bulk-to-surface migration of oxygen. As confirmed by the  $\text{H}_2$ -1st and 2nd TPR results (Table 2), AgCeNd<sub>0.05</sub>-A and especially AgCe-A generates less bulk oxygen-derived  $\text{O}_x^-$  species compared with the corresponding fresh catalysts, while there are no obvious differences between AgCeNd<sub>0.1</sub>-A and AgCeNd<sub>0.1</sub>-F (Fig. 8c and f).

On all accounts, sintering of Ag reduces the catalyst availability for active oxygen (both from gaseous  $\text{O}_2$  and ceria bulk oxygen), which thus reduces the soot oxidation activity of AgCe-A drastically. Meanwhile, Nd doping restrains Ag particle growth and hereby improves the catalyst stability towards thermal ageing.

Second, about the chemical states of Ag: as indicated by the XPS results (Fig. 6b and c), AgCe-A exhibits slightly higher content of  $\text{Ag}_2\text{O}$  than AgCeNd<sub>0.05</sub>-A and AgCeNd<sub>0.1</sub>-A, which may be attributed to more remarkable oxygen reverse spillover on Ag/ $\text{CeO}_2$  as mentioned above. Anyhow, these differences in Ag chemical states should not be crucial for the catalytic reaction, the reasons are twofold: On one hand, the amount of these  $\text{Ag}_2\text{O}$  is very low (they cannot be detected either by XRD or HRTEM). Furthermore, AgCe-A with higher content of  $\text{Ag}_2\text{O}$  shows no more  $\text{H}_2$  consumption than AgCeNd<sub>0.05</sub>-A (see  $\text{O}_2$ -1st in Fig. 8d and e), which also implies a limited contribution of the surface  $\text{Ag}_2\text{O}$  species to the catalytic reaction. On the other hand, as evidenced by Haneda [43] and Corro et al. [44], during soot oxidation,  $\text{Ag}_2\text{O}$  can act as the oxidizing agent which gets reduced readily by soot to form metallic Ag. Therefore, it is also impossible for the  $\text{Ag}_2\text{O}$  species to persist and do continuous influence on the reaction. In a word, the differences in silver chemical states cannot determine the catalytic cycle.

Finally, about the concentration of oxygen vacancies: ageing treatment with 1%  $\text{O}_2$  led to oxidation of  $\text{Ce}^{3+}$  to  $\text{Ce}^{4+}$  over all the catalysts (Fig. 6c and d). With reduction of  $\text{V}_{\text{O-s}}$ , AgCeNd<sub>0.05</sub>-A and AgCeNd<sub>0.1</sub>-A generated higher amounts of  $\text{O}_2^-$  during  $\text{O}_2$ -1st, 2nd and 3rd tests than corresponding fresh catalysts (Table 2) [5,10]. Separately, decrease in  $\text{V}_{\text{O-b}}$  content hamper the utility of ceria bulk oxygen over AgCeNd<sub>0.05</sub>-A obviously (comparing the  $\text{H}_2$ -1st results in Fig. 8b and e). As for AgCe-A, the influences of Ag sintering and  $\text{V}_{\text{O-b}}$  reduction overwhelms that of  $\text{V}_{\text{O-s}}$  decrease, making this catalyst hardly available for active oxygen and the least active among the aged catalysts.

As a summary, we consider the changes in Ag particle sizes are the most important factors that determine the catalytic behavior of the aged catalysts. Nd doping inhibits the growth of Ag, and thus improves the catalyst stability at high temperatures. Although reduction in  $V_{O-s}$  may benefit the catalytic performance, its influences can be shadowed by Ag sintering. Consequently, AgCe-A with severe Ag sintering is almost inert for soot oxidation. Meanwhile, AgCeNd<sub>0.05</sub>-A suffered from decrease in bulk-to-surface oxygen migration caused by  $V_{O-b}$  reduction. As for AgCeNd<sub>0.1</sub>-A, it exhibits reduced  $V_{O-s}$  amount, increased  $V_{O-b}$  content and unchanged Ag particle sizes comparing with AgCeNd<sub>0.1</sub>-F, and thus shows no deactivation even after ageing (see Fig. 7b).

## 5. Conclusions

In this study, by characterization and comparison of three cubic Ag/Ce<sub>x</sub>Nd<sub>1-x</sub>O<sub>2</sub> catalysts, several conclusions can be drawn:

- Both activation of gaseous O<sub>2</sub> and oxygen bulk-to-surface migration were confirmed to play important roles in soot catalytic oxidation over Ag/CeO<sub>2</sub> catalysts. Any changes in catalyst structure may affect these two factors, and thus alter the catalyst performance.
- Nd doping increases surface oxygen vacancies ( $V_{O-s}$ ), but decreases bulk oxygen vacancies ( $V_{O-b}$ ) over Ag/CeO<sub>2</sub>. This suppresses the utilization of ceria bulk oxygen, and induces the formation of less active O<sub>2</sub><sup>-</sup> species instead of O<sub>2</sub><sup>•-</sup>, resulting in reduction of the soot oxidation activity.
- The surface oxygen vacancies and mesopores in the ceria support generated via Nd doping inhibit Ag sintering at high temperatures, which hereby improves the catalyst stability towards high-temperature ageing effectively.

Based on these conclusions, it is suggested that cubic Ag/CeO<sub>2</sub> catalysts can be more practical with the introduction of Nd. However, the Nd doping amount should be controlled moderately to ensure both the activity and thermal stability of Ag/Ce<sub>x</sub>Nd<sub>1-x</sub>O<sub>2</sub>. Under the specific conditions applied in this work, the optimal Nd loading is 5%.

## Acknowledgements

The authors would like to acknowledge Project 2015AA034603 by the Ministry of Science and Technology of China, China Postdoctoral Science Foundation (2015M580607), Project 113007A by the Ministry of Education of China and the Fundamental Research Funds for the Central Universities. Moreover, we would also thank the financial support from the Key Laboratory of Advanced Materials of Ministry of Education of China.

## Appendix A. Supplementary data

Supplementary data associated with this article can be found, in the online version, at <http://dx.doi.org/10.1016/j.apcatb.2016.10.006>.

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